



Refractive index dispersion and determination of the optical constants of BaTiO₃ thin film prepared by sol gel

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The optical properties of the BaTiO₃ film deposited on ITO coated glass by sol–gel method using barium acetate and titanium isopropoxide as starting materials have been investigated. The optical constants such as refractive index absorption coefficient and dielectric constant were determined using optical spectra. Type of the optical transition was found to be a direct transition with an E_g value of 3.47 eV. The refractive index dispersion of the film obeys to single oscillator model with dispersion energy of 21.07 eV, oscillator energy of 5.20 eV and dispersion parameter E_0/S_0 of 7.3×10^{-14} eV.m². The obtained results indicate that the BaTiO₃ film can be a probable candidate for optoelectronic applications as a wide gap material.

Keywords: BaTiO₃ thin film, AFM, Optical constant, Refractive index, Sol gel.

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1. Introduction

C Metal oxide with perovskite structure such as BaTiO₃ and Pb(Zr,Ti)O₃ are widely used in various electronic applications such as multi-layer ceramic capacitors (MLCC), transducers, actuators and ferroelectric random access memories (FRAM) due to their unique ferroelectric, pyroelectric and piezoelectric properties [1–4]. Barium titanate is one of the most widely studied ferroelectric materials due to its many potential useful properties and it widely applied in electronic devices as high permittivity capacitors, infrared detectors or transducers. It is well known that ferroelectric and optical properties of ferroelectric ceramics depend on the grain size [5]. The properties of BaTiO₃ thin films change with various deposition parameters like deposition technique, substrate temperature and substrate type. The function of BaTiO₃ thin films in optoelectronic devices depends on structural and optical properties like grain size, lattice distortion, micro-strain, band gap, refractive index and absorption [6]. Thus,

it is important to study how optical and structural properties depend on the deposition parameters.

With this aim, we have studied structural properties and the optical characteristics and of BaTiO₃ thin film deposited onto indium tin oxide coated glass substrate by sol gel method.

2. Experimental

For preparation of BaTiO₃ film, the used starting materials are barium acetate and titanium isopropoxide and glacial acetic acid were used as starting materials. Firstly, 1.66 M Ba(CH₃COO)₂ was dissolved in 70 ml of glacial acetic acid and stirred at 50 oC for 30 min and then, it was cooled down to the room temperature. The stoichiometric amount of titanium isopropoxide was dissolved in 70 ml isopropanol and stirred by magnetic stirrer. After dissolution of two compounds, the precursor solution was mixed with a solution consisting of Ba(CH₃COO)₂ and titanium isopropoxide and it was stirred for 1 h. After stirring, the

solution was refluxing for 2 h. The BaTiO₃ solution was coated onto indium tin oxide coated glass with 2000 rpm using a VTC-50 DeskTop Spin Coater. The obtained films were dried at 60 oC for 10 min and it was annealed at 650 oC for 1h. The transmittance, absorbance and reflectance spectra of the film were measured using a Shimadzu UV-VIS-NIR 3600 spectrophotometer.

3. Results and Discussion

3.1. Determination of optical constants of the BaTiO₃ thin film

The atomic force microscopy image of the BaTiO₃ film is shown in Fig.1. The film was not formed from a certain particles. The film has a homogeneous surface with a surface roughness of 16.77 nm.

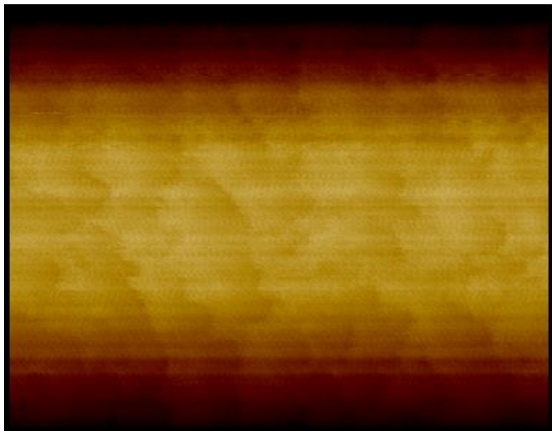


Fig.1. AFM image of the BaTiO₃ film on ITO glass

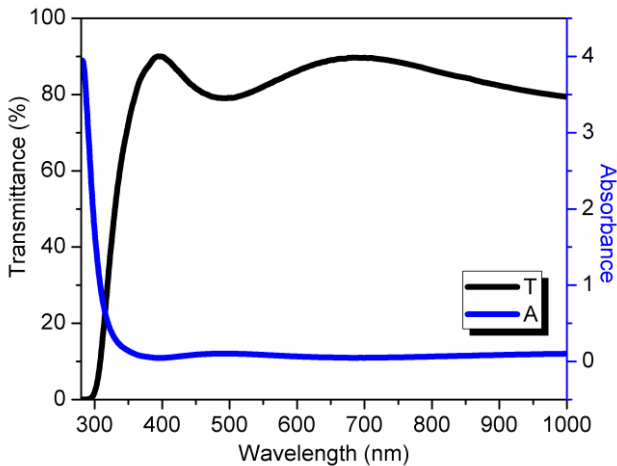


Fig. 2. Absorbance and transmittance spectra of the BaTiO₃ thin film.

Fig.2 show the absorbance (A) and transmittance (T) spectra of the BaTiO₃ thin film. Transmittance (T), absorption coefficient (α) and reflectance (R) are related through the following relations [7-8]:

$$T(\lambda) = \frac{(1 - R(\lambda))^2 e^{-\alpha d}}{1 - R(\lambda)^2 e^{-2\alpha d}}, \quad (1)$$

where d is the film thickness, and the reflectance which is described as [7]:

$$R(\lambda) = \frac{(n(\lambda) - 1)^2 + k(\lambda)^2}{(n(\lambda) + 1)^2 + k(\lambda)^2}, \quad (2)$$

The solution of this Eq.2 can be given as:

$$n(\lambda) = \frac{(1 + R(\lambda)) + \sqrt{4R(\lambda) - (1 - R(\lambda))^2 k(\lambda)^2}}{1 - R(\lambda)}, \quad (3)$$

where k is the extinction coefficient given by [7]:

$$k = \frac{\alpha \lambda}{4\pi}, \quad (4)$$

Thus, the refractive index and the extinction coefficient can be calculated by means of these equations. The refractive index of any material is an important parameter for optical communications. It is very important to analyze refractive index dispersion of the investigated materials. The complex refractive index can be described by the following relation:

$$\hat{n}(E) = n(E) + ik(E), \quad (5)$$

where $n(E)$ is the real part of the refractive index and $k(E)$ is the imaginary part of the refractive index also called the extinction coefficient. The $n(E)$ and $k(E)$ constants can be determined from the following relations [9]:

$$n(E) = 1 + \frac{2}{\pi} \int_0^{\infty} \frac{E' k(E')}{(E')^2 - E^2} dE', \quad (6)$$

$$k(E) = -\frac{2}{\pi} \int_0^{\infty} \frac{n(E') - 1}{(E')^2 - E^2} dE', \quad (7)$$

The refractive index of the BaTiO₃ thin film was calculated using Eq. 3 and the spectral dependence of the refractive index is shown in Fig. 3. It is clear that the refractive index was decreased with increasing the wavelengths, i.e, it exhibits a normal dispersion. The dispersion behavior of the BaTiO₃ thin film can be analyzed by Wemple and DiDomenico given by the following relation [10-11],

$$n^2 - 1 = \frac{E_o E_d}{E_o^2 - (hc/\lambda)^2}, \quad (8)$$

where E_o is the single oscillator energy and E_d is the dispersion energy, which is a measure of the intensity of the inter band optical transitions. The plot of $(n^2 - 1)^{-1}$ vs E^2 is shown in Fig.4 and E_o and E_d values were determined from the slope, $(E_o E_d)^{-1}$ and intercept (E_o/E_d) , on the vertical axis

and were found to be 5.20 eV and 21.07 eV, respectively. The relation between E_o and E_g for the BaTiO₃ was found to be $E_o=1.5 E_g$. This average gap gives quantitative information on the overall band structure of the BaTiO₃. The obtained E_o value is in agreement with the value of melt grown BaTiO₃ [12] and it is smaller than that of RF sputtered strontium substituted barium titanate thin films [13]. This difference is due to the substituted strontium into BaTiO₃.

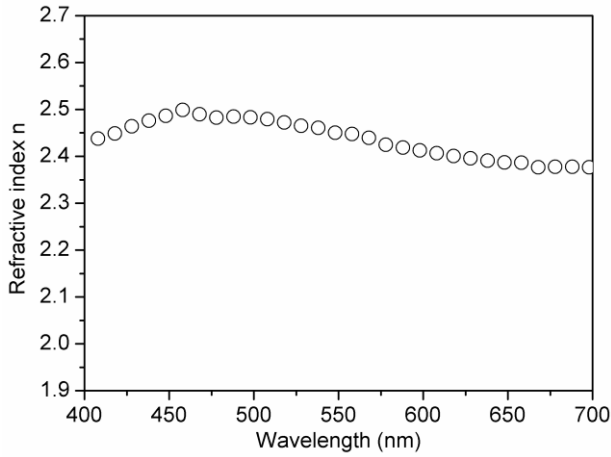


Fig. 3. Refractive index as a function of wavelength of the BaTiO₃ thin film.

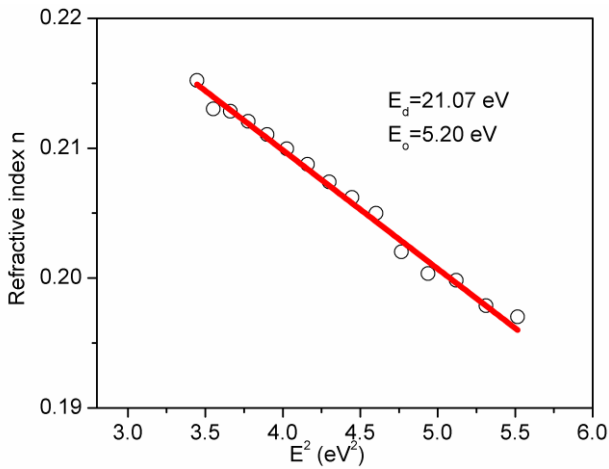


Fig. 4. Plot of $(n^2-1)^{-1}$ vs E^2 of the BaTiO₃ thin film.

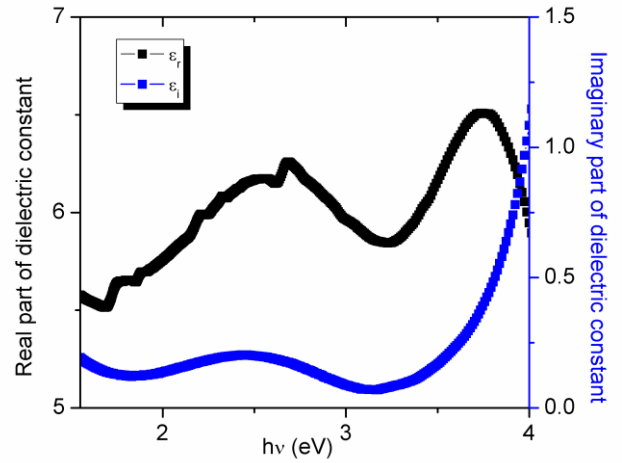


Fig. 5. Real and imaginary parts dependence on the photon energy of the BaTiO₃ thin film.

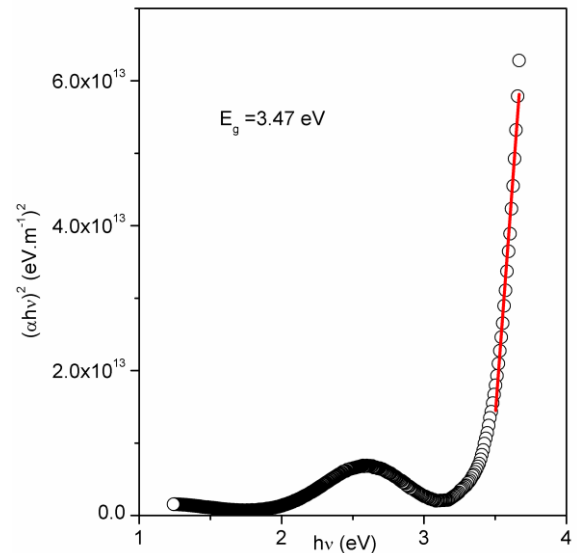


Fig. 6. Plot of $(\alpha hv)^2$ v.s $h\nu$ of the BaTiO₃ thin film

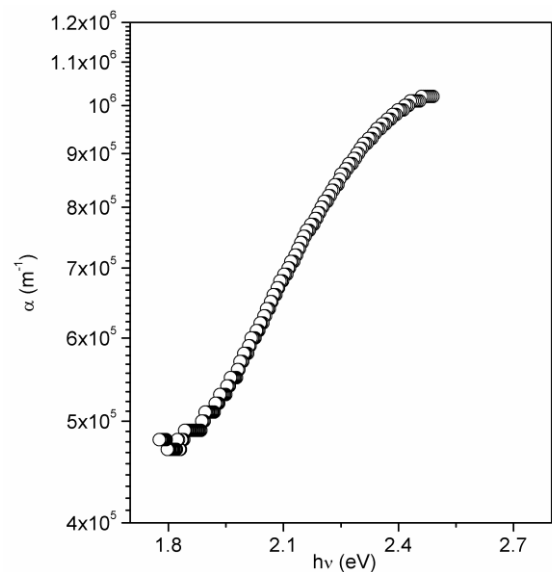


Fig. 7. Plot of $\log \alpha$ vs $h\nu$ of the BaTiO₃ thin film

The dispersion energy is related to other physical parameters of the BaTiO₃ material and it can be expressed by the following relation [10]:

$$E_d = \beta N_c Z_a N_e, \quad (9)$$

Where Z_a is the formal chemical valency of the anion (2 for BaTiO₃), N_c is the effective coordination number of the cation nearest neighbour to the anion (6 for BaTiO₃), N_e is the effective number of valence electrons per anion (8 for BaTiO₃). The β value for the BaTiO₃ film was found to be 0.22. This suggests that the BaTiO₃ has the ionic bonding. The Sellmeier single oscillator relation can be defined as follows:

$$n^2 - 1 = \frac{S_o \lambda_o^2}{1 - (\lambda_o / \lambda)^2}, \quad (10)$$

where λ is the wavelength of incident light, S_o is the average oscillator strength and λ_o is an average oscillator wavelength. The relation between S_o and n_∞ values can be given by the following relation:

$$\frac{n_\infty^2 - 1}{n^2 - 1} = 1 - \left(\frac{\lambda_o}{\lambda} \right)^2, \quad (11)$$

Thus, the average oscillator strength can be obtained by the following relation:

$$S_o = \frac{n_\infty^2 - 1}{\lambda_o^2} \quad (12)$$

Where n_∞ is the long wavelength refractive index. In order to obtain S_o , n_∞ and λ_o values, we plotted the curve of $n^2 - 1$ vs λ^2 and the S_o and λ_o values were found to be $7.11 \times 10^{13} \text{ m}^{-2}$, 2.24 and 238.84 nm, respectively. The dispersion parameter E_o/S_o for the studied BaTiO₃ was found to be $7.3 \times 10^{-14} \text{ eV} \cdot \text{m}^2$ and this value is in agreement with that of melt grown BaTiO₃ sample [12]. This suggests that the dispersion of the refractive index for the perovskite oxides is identical. The M_{-1} and M_{-3} moments of the optical spectra can be expressed as [14]:

$$E_o^2 = \frac{M_{-1}}{M_{-3}}, \quad E_d^2 = \frac{M_{-1}^3}{M_{-3}}, \quad (13)$$

The M_{-1} and M_{-3} moments were determined using Eq. 12 and were obtained to be 4.05 and 0.15 eV⁻². The complex dielectric constant of the film is expressed as [7]:

$$\varepsilon^* = \varepsilon_1(\omega) + i\varepsilon_2(\omega) = n^2(\omega) - k^2(\omega) + 2in(\omega)k(\omega) \quad (14)$$

where ε_1 is the real part and ε_2 is the imaginary of the dielectric constant. Fig. 5 shows the real and imaginary parts dependence on photon energy. The values of ε_1 and ε_2 increase with increasing the photon energy. The real part of the dielectric constant is higher than imaginary part of the dielectric constant. The real part spectrum of the dielectric constant shows two peaks. One of them at 3.75 eV corresponds to the absorption edge and the presence of this peak is due to electronic transitions between valence and

conduction bands. The other peak at 2.68 eV is due to electronic transitions taken place in intrabands.

3.2. Determination of the optical band gap of BaTiO₃ film

It is well-known that the optical absorption spectrum is an important tool to obtain optical band gap of crystalline and amorphous materials. The band structure of the BaTiO₃ obeys the rule of direct transition. The optical transitions occurring in semiconductors can be explained as follows. For indirect optical transitions; the absorption coefficient α_i for indirect transition is expressed by the following relation:

$$\alpha_i = \alpha_e + \alpha_a, \quad (15)$$

where α_e is the contribution of emission and α_a is the contribution of absorption of phonons. Thus, α_i can be written as [15]:

$$(\alpha_i h\nu) = \frac{[h\nu - (E_g - E_p)]^2}{[\exp(E_p / kT) - 1]} + \frac{[h\nu - (E_g + E_p)]^2}{[1 - \exp(-E_p / kT)]} \quad (16)$$

Absorption coefficient due to phonon absorption and emission is expressed as:

$$\alpha_a = A(T) \frac{[h\nu - (E_g - E_p)]^2}{h\nu}, \quad h\nu > E_g - E_p \quad (17)$$

$$\alpha_e = B(T) \frac{[h\nu - (E_g + E_p)]^2}{h\nu}, \quad h\nu > E_g + E_p \quad (18)$$

where $A(T)$ and $B(T)$ are constants. For indirect transitions, the plot of $(\alpha h\nu)^{1/2}$ vs $h\nu$ at lower photon energies, the straight line corresponds to the phonon absorption and its intercept is equal to $E_g - E_p$ and at higher photon energies, the other straight line corresponds to the phonon emission process and its intercept equals to $E_g + E_p$. In direct transitions, the absorption coefficient α_d for direct transition is expressed by the following relation:

$$\alpha_d = C(T) \frac{[h\nu - E_p]^{1/2}}{h\nu}, \quad (19)$$

where $C(T)$ is constant. For the direct transitions, the plot of $(\alpha h\nu)^{1/2}$ vs $h\nu$ indicates a straight line. We have analyzed in detail to determine optical transitions between valence band and conduction bands. It is evaluated that the BaTiO₃ has a direct optical band gap. Thus, the curve of $(\alpha h\nu)^2$ vs $h\nu$ was plotted and is shown in Fig. 6 and this plot shows one straight line. The E_g value of the film was determined from

this linear part and was found to be 3.47 eV. The obtained E_g value is in good agreement with the band gap energy of 3.47 eV obtained by Suzuki et al [16] for BaTiO₃ thin film prepared by radio-frequency-plasma. The obtained E_g value of the studied BaTiO₃ film is close to the data of the films obtained, But it is smaller than that of other films obtained with metallo-organic solution method (3.5 eV) [17], sol-gel technique (3.53 eV) [18], RF sputtering (3.53–3.68 eV), BaTiO₃ powders (3.92 eV) and sol-gel derived BaTiO₃ films on ITO coated glass [1,19-21]. This suggests that the optical band gap of the BaTiO₃ film changes with preparation method and production conditions.

At lower photon energies, the absorption behavior of the BaTiO₃ film can be analyzed by following relation:

$$\alpha \propto \exp(E/E_u), \quad (20)$$

Where E_u is Urbach energy, which is interpreted as the width of the tails of localized states in the band gap [22]. In Urbach region, the optical transitions are occurred between extended states in one band and localized states in the exponential tail of the other band. To obtain Urbach energy E_u , we plotted the curve of $\log \alpha$ vs $h\nu$, as shown in Fig. 7. The E_u value was determined from the slope of Fig. 7 and was found to be 308±2 meV.

4. Conclusion

The BaTiO₃ film was prepared by sol-gel method. The optical constants such as refractive index absorption coefficient and dielectric constant were determined using optical spectra. The refractive index dispersion behavior was analyzed by single oscillator model and the dispersion and oscillator parameters were determined which in good agreement with other studies. The obtained optical parameters indicate that the BaTiO₃ film can be a probable candidate for optoelectronic applications.

References

- [1] H.X. Zhang, C.H. Kam, Y. Zhou, X.Q. Han, Y.L. Lam, Y.C. Chan, K. Pita, *Materials Chemistry and Physics* 63 (2000) 174–177.
- [2] S. Mathews, R. Ramesh, T. Venkatesan, J. Benedetto, *Science* 276 (1997) 238.
- [3] J.F. Scott, *Ferroelectr. Rev.* 1 (1998) 1.
- [4] B.H. Park, B.S. Kang, S.D. Bu, T.W. Noh, J. Lee, W. Jo, *Nature* 401 (1999) 682.
- [5] H.S. NALWA, *Handbook of Low and High Dielectric Constant Materials and Their Applications*, USA, Academic Press, 1999.
- [6] Y. K. Vayunandana Reddy, D. Mergel, S. Reuter, V. Buck, and M. Sulkowski, *J. Phys. D: Appl. Phys.* 39 (2006) 1161–1168.
- [7] T.S. Moss, G.J. Burell, B. Ellis, *Semiconductor Opto-Electronics*, Butterworth, London (1973).
- [8] B.G. Yacobi, *Semiconductor Materials: An Introduction to Basic Principles*, Kluwer Academic Publishers, New York, Boston, Dordrecht, London, Moscow, (2002)
- [9] H.A. Kramers, *J. Opt. Soc. Am.* 12 (1926) 547.
- [10] S.H. Wemple, M. DiDomenico, *Phy. Rev B.* 3 (1971) 1338.
- [11] F. Yakuphanoglu, A. Cukurovali, İ. Yilmaz, *Physica B* 353 (2004) 210–216
- [12] S. H. Wemple, M. DiDomenico, Jr. and I. Camlibel, *J. Phys. Chem. Solids* 29 (1968) 1797-1803.
- [13] B. Panda, A. Dhar, G.D. Nigam, D. Bhattacharya, S.K. Ray, *Thin Solid Films* 332 (1998) 46-49
- [14] S.H. Wemple, M. DiDomenico, *Phys. Rev. Lett.* 23 (1969) 1156
- [15] G.G. Macfarlane, V. Roberts, *Phys. Rev.* 97 (1955) 1714
- [16] C.W. Teng, J.F. Muth, U. Ozgur, M.J. Bergman, H.O. Everitt, A.K. Sharma, C. Jin, J. Narayan, *Appl. Phys. Lett.* 76 (2000) 979.
- [17] P.C. Joshi, S.B. Desu, *Thin Solid Films* 300 (1997) 289.
- [18] M.N. Kamalasanan, N.D. Kumar, S. Chandra, *J. Appl. Phys.* 76 (1994) 4603.
- [19] J.S. Zhu, X.M. Lu, W. Jiang, W. Tian, M. Zhu, M.S. Zhang, X.B. Chen, X. Liu, Y.N. Wang, *J. Appl. Phys.* 81 (1997) 1392.
- [20] A. Mansingh, C.V.R. Vasanta Kumar, *J. Mater. Sci. Lett.* 7 (1988) 1104.
- [21] A. Lanculescu, M. Gartner, B. Despax, V. Bley, Th. Lebey, R. Gavrilă, M. Modreanu, *Applied Surface Science* 253 (2006) 344–348
- [22] J.C. Tauc, *Optical Properties of Solids*, North-Holland, Amsterdam, 1972.